

The Nb-Ni-Ti (Niobium-Nickel-Titanium) System—Update

K.P. Gupta, The Indian Institute of Metals

Nb-Ni-Ti system was reviewed by [1991Gup] on the basis of experimental data available till 1988. Recently, a complete isothermal section of the Nb-Ni-Ti system at 900 °C was established and is reported here.

$\beta\text{Ti} \rightarrow \text{cph } \alpha\text{Ti}$ transformation occurs at the Ti side at $T < 765$ °C.

Binary Systems

The binary systems Nb-Ni, Nb-Ti, and Ni-Ti were reviewed earlier, and the accepted phase diagrams are those given by [Massalski2] (Fig. 1-3). For the purpose of discussing the new results of the Nb-Ni-Ti system, it is relevant to mention the binary phases that exist in the three binary systems at 900 °C. The Nb-Ni system has three intermediate phases: the Nb_7Ni_6 (μ), NbNi_3 (π), and NbNi_8 (δ); the δ phase exists at $T < 535$ °C. Three intermediate phases exist in the Ni-Ti system: Ni_3Ti (ρ), NiTi (β), and NiTi_2 (ζ); all three phases exist at 900 °C. The Nb-Ti system is an isomorphous system with Nb and Ti forming a body-centered cubic (bcc) solid solution α at all compositions. The bcc

Binary and Ternary Phases

In the review of the Nb-Ni-Ti system [1991Gup] used the available experimental data of [1958Kor], [1966Vuc], and [1966Pry]. While [1958Kor] and [1966Vuc] studied the NbNi_3 - Ni_3Ti section of the Nb-Ni-Ti system, [1966Pry] studied the Ni-rich region (up to 50 at.% Nb and 50 at.% Ti) of the ternary system and established two partial isothermal sections at 1000 and 900 °C. [1958Kor], using thermal analysis, metallography, electrical resistivity, and hardness measurements on a series of $\text{Ni}_3(\text{Nb},\text{Ti})$ alloys, suggested the existence of an isomorphous system with a minimum of 1285 °C at ~19 at.% Ti. An isomorphous pseudo-binary in NbNi_3 - Ni_3Ti system is, however, not possible because the two terminal phases are not isostructural. A probable peritectic-type pseudo-binary that needs experimental verification was suggested by [1991Gup] for the NbNi_3 - Ni_3Ti system. The ternary intermediate phase reported by [1966Vuc] and [1966Pry] are given in Table 1.

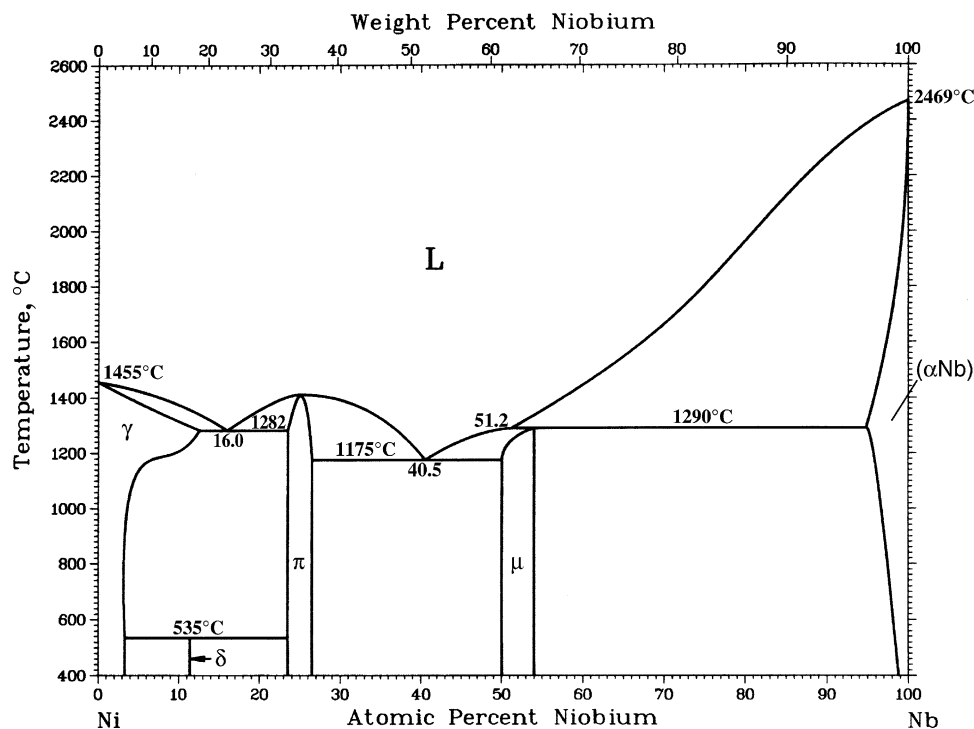


Fig. 1 Binary Nb-Ni system [Massalski2]

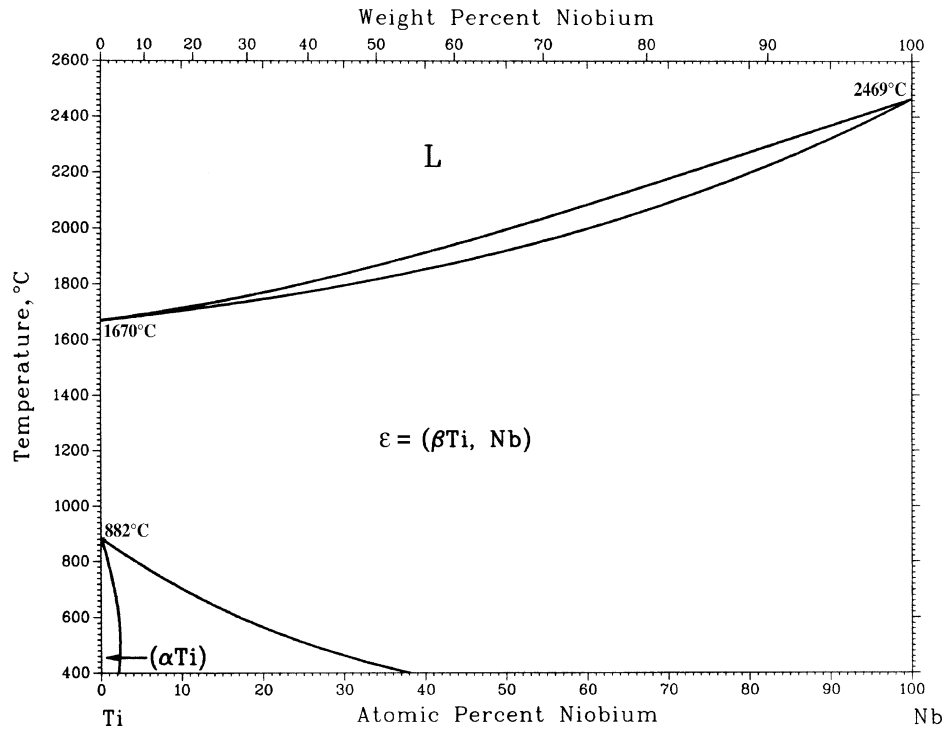


Fig. 2 Binary Nb-Ti system [Massalski2]

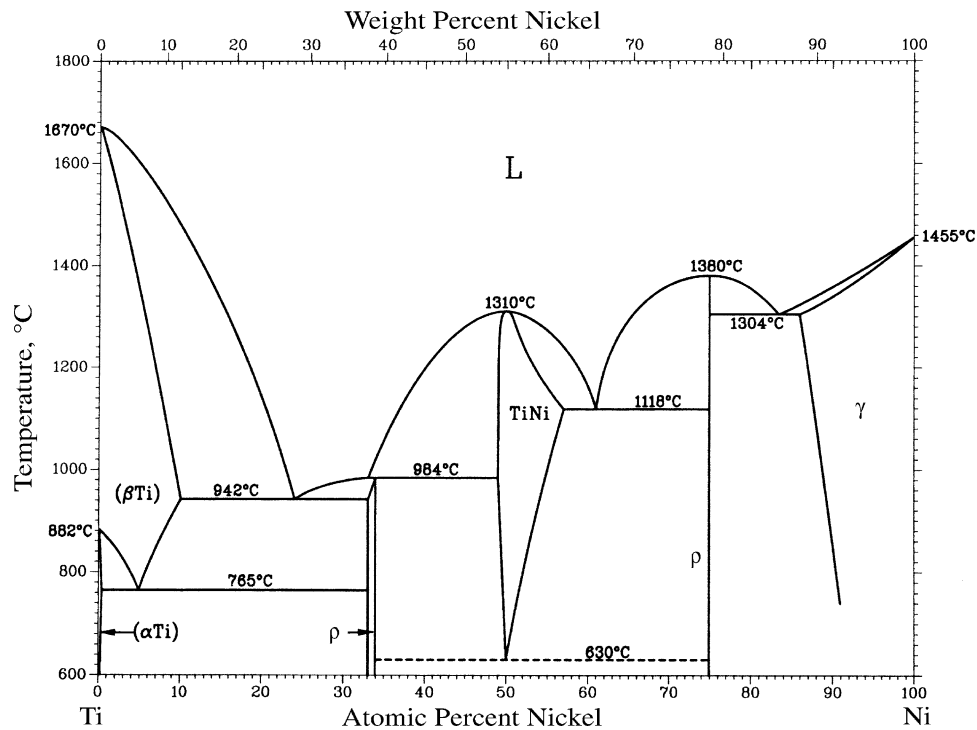


Fig. 3 Binary Ni-Ti system [Massalski2]

Ternary System

The Nb-Ni-Ti system has been studied recently by [2000Gua] at 900 °C using a diffusion couple technique.

For this study five binary alloys— $\text{Ti}_{80}\text{Ni}_{20}$ (TN_2), $\text{Ti}_{70}\text{Ni}_{30}$ (TN_3), TiNi, $\text{Nb}_{25}\text{Ni}_{75}$ (Ni_3Nb), and (Ni_6Nb_7) —were arc melted using 99.99 mass-% pure component elements. Slices of these alloys and of pure Ni and Ti were prepared

Section II: Phase Diagram Evaluations

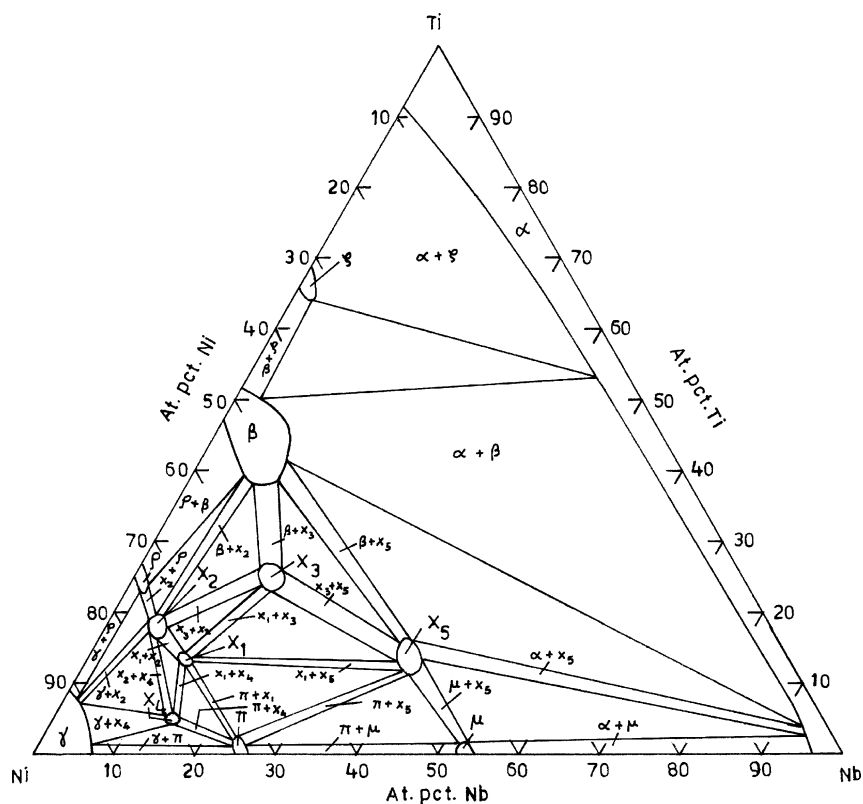


Fig. 4 The 900 °C isothermal section of the Nb-Ni-Ti system [2000Gua]

Table 1 Phases present in the Nb-Ni-Ti system and their structure data

Phase designation	Composition	Pearson's symbol	Space group	Type	Lattice parameter, nm		
					<i>a</i>	<i>b</i>	<i>c</i>
α	(β Ti,Ni), (β Ti,Nb)	<i>cI2</i>	<i>Im</i> $\bar{3}m$	W
γ	(Ni)	<i>cF4</i>	<i>Fm</i> $\bar{3}m$	Cu
ε	(α Ti,Ni), (α Ti,Nb)	<i>hP2</i>	<i>P6</i> $\bar{3}/mmc$	Mg
δ	NbNi ₈	<i>tI18</i>	...	NbNi ₈	1.08	...	0.36(a)
π	NbNi ₃	<i>oP8</i>	<i>Pmmm</i>	Cu ₃ Ti	0.5116	0.4259	0.4565
μ	Nb ₇ Ni ₆	<i>hR13</i>	<i>R</i> $\bar{3}m$	W ₆ Fe ₇	0.4893	...	2.664(b)
ρ	Ni ₃ Ti	<i>hP16</i>	<i>P6</i> $\bar{3}/mmc$	Ni ₃ Ti	0.5101	...	0.8307
β	NiTi	<i>cP2</i>	<i>Pm</i> $\bar{3}m$	CsCl	0.3015
ζ	NiTi ₂	<i>cF96</i>	<i>Fd</i> $\bar{3}m$	CFe ₃ W ₃	1.1324
X ₁ (c)	Nb ₁₃ Ni ₇₅ Ti ₁₂	<i>h</i>	0.257	...	0.422
X ₂	Nb ₅ Ni ₇₅ Ti ₂₀
X ₃	Nb ₁₅ Ni ₅₆ Ti ₂₉	<i>o</i>	0.879	1.187	0.881
X ₄	Nb ₁₅ Ni ₈₀ Ti ₅
X ₅	Nb _{40.3} Ni _{45.1} Ti _{14.6}
Z ₁ (d)	Nb _{0.75} Ni ₇₅ Ti _{24.25}	<i>h</i>	0.5106	...	2.081
Z ₂	Nb _{2.75} Ni ₇₅ Ti _{22.25}	<i>hR</i>	...	BaPb ₃	0.5118	...	1.8809
Z ₃	Ni _{8.25} Ni ₇₅ Ti _{16.75}	<i>h</i>	...	Ni ₃ Sn	0.5130	...	0.4211(e)

(a) Lattice parameter of face-centered tetragonal (fct) cell with 36 atoms/cell. (b) Lattice parameter of hexagonal cell with 39 atoms/cell. (c) X_{*n*} with *n* = 1-4 from [1966Pry] and X₅ from [2000Gua]. (d) Z_{*n*} phases are from [1966Vuc]; Z₁ and Z₂ existed only in the as-cast state of the alloys. (e) Probably the same phase as X₁ with doubled *a* parameter

by spark erosion, mechanically polished, and resistance welded in vacuum to form the following diffusion couples: (TiNi-Nb₆Ni₇)/Ni, (TiNi-Ni₃Nb)/Ni, (TiNi-TN₃)/Ni, (Ti-TN₂)/TiNi, (TiNi-TN₃)/Ni, and (Ti-TN₃)/TiNi. The diffusion couples, sealed in evacuated silica capsules, were annealed for 400 h at 900 °C and water quenched. The diffusion zones of the various couples were examined metallographically, and an electron probe microanalyzer (EPMA) was used to determine the concentration penetration curves for each element to identify the phases present in the diffusion zones and to determine the tie lines of phases in equilibrium. The 900 °C isothermal section by [2000Gua] is given in Fig. 4.

Figure 4 shows the presence of all the binary phases existing at 900 °C, that is, the ρ , β , ζ , π , and μ phases. Extension of all the binary phases, except the β phase, is <5 at.% Nb or Ti. The β phase extends up to ~10 at.% Nb. A small face-centered cubic (fcc) γ -phase region exists at the Ni corner. The bcc α phase region extends from the Nb corner to the Ti corner; the solubility of Ni in the α phase is smaller at the Nb side than at the Ti side. In this investigation the presence of a new ternary phase X₅ was found between the Nb₇Ni₆ and NiTi phase at ~Nb₄₀Ni₄₅Ti₁₅ composition. The other intermediate phases found in this investigation are X₁, X₂, and X₃, which were reported earlier by [1966Pry]. The X₄ phase region was not found due to the choice of diffusion couples, and it is shown in Fig. 4 on the basis of the results of [1966Pry]. The new ternary intermediate phase X₅ is in equilibrium with the α , μ , β , X₃, X₁, and π . The bcc α phase is found

in equilibrium with the ζ , β , X₅, and μ phases. The X₃ phase is in equilibrium with the β , X₂, X₁, and X₅ phases. The X₂ phase is in equilibrium with the fcc γ , ρ , X₄, X₁, and X₃ phases. The X₁ phase is in equilibrium with the X₂, X₄, π , X₅, and X₃ phases. The X₄ phase is in equilibrium with the γ , X₂, X₁, and π phases. Whether the X₅ phase also exists at 1000 °C is unknown. The crystal structures of the phases were not determined by [2000Gua].

References

- 1958Kor:** I.I. Kornilov and E.N. Pylaeva, The Binary Systems Ni₃Ti-Ni₃Ta, Ni₃Ti-Ni₃Nb, *Zh. Neorg. Khim.*, 1958, **3**, p 673-677 (in Russian) (Phase Equilibria, #)
- 1966Pry:** L.I. Pryakhina, K.P. Myasnikova, V.V. Burnashova, E.E. Cherkashin, and V.Ya. Markiv, Ternary Intermetallic Compounds in the System Ni-Ti-Nb, *Poroshk. Metall.*, 1966, **8**(44), p 61-69 (in Russian) (Phase Equilibria, #)
- 1966Vuc:** J.H.N. Van Vucht, Influence of Radius Ratio on the Structure of Intermetallic Compounds of the AB₃ Type, *J. Less Common Met.*, 1966, **11**, p 308-322 (Crys Structure)
- 1991Gup:** K.P. Gupta, *Phase Diagrams of Ternary Nickel Alloys, Part 2*. The Indian Institute of Metals, Calcutta, 1991 (Review)
- 2000Gua:** Y. Guanjun and H. Shiming, Study of the Phase Equilibria of the Ti-Ni-Nb Ternary System at 900 °C, *J. Alloys Compds.*, 2000, **297**, p 226-230 (Phase Equilibria, #)

indicates presence of phase diagram.

Nb-Ni-Ti evaluation contributed by **K.P. Gupta**, The Indian Institute of Metals, Metal House, Plot 13/4, Block AQ, Sector V, Calcutta, India. Literature searched through 2000. Dr. Gupta is the Alloy Phase Diagram Co-Category Program Editor for ternary nickel alloys.